

Reducing Facet Nucleation during Algorithmic Self-Assembly

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ABSTRACT

Algorithmic self-assembly, a generalization of crystal growth, has been proposed as a mechanism for bottom-up fabrication of complex nanostructures and autonomous DNA computation. In principle, growth can be programmed by designing a set of molecular tiles with binding interactions that enforce assembly rules. In practice, however, errors during assembly cause undesired products, drastically reducing yields. Here we provide experimental evidence that assembly can be made more robust to errors by adding redundant tiles that “proofread” assembly. We construct DNA tile sets for two methods, uniform and snaked proofreading. While both tile sets are predicted to reduce errors during growth, the snaked proofreading tile set is also designed to reduce nucleation errors on crystal facets. Using atomic force microscopy to image growth of proofreading tiles on ribbon-like crystals presenting long facets, we show that under the physical conditions we studied the rate of facet nucleation is 4-fold smaller for snaked proofreading tile sets than for uniform proofreading tile sets.

Molecular self-assembly is an emerging technology that will ultimately enable the fabrication of great quantities of complex nanoscale objects such as computer circuits at very low cost. The difficulty of self-assembling complex objects lies in the fact that precise direction of assembly in the form of a set of instructions (an algorithm) must be encoded in the molecules themselves. Nucleic acids allow the construction of programmable units for self-assembly,¹ here called “tiles”, that have been used to create periodic lattices^{2–4} and small finite assemblies.^{5–7} Algorithmic self-assembly, a generalized form of crystal growth,⁸ has been proposed as a general method for directing the growth of complex finite and infinite objects.⁹ In principle, arbitrarily complex objects can be constructed using algorithmic self-assembly,¹⁰ and objects including large finite-sized shapes and some circuit diagrams can be assembled from a small number of components.^{11–14} Aperiodic one-^{15,16} and two-dimensional^{17,18} structures have been algorithmically self-assembled from programmable crystal monomers constructed from DNA tiles.

In algorithmic self-assembly, crystal growth is programmed by designing a set of tiles with binding interactions

that enforce specific local assembly rules. Growth begins from a nucleating structure and consists of a series of attachments of single tiles. Under slightly supersaturated conditions, the attachment of a tile to a growing crystal is energetically favorable only if it attaches to a growing crystal by at least two binding sites. The tiles are designed so that during correct assembly, at every step a tile in the pattern attaches by a particular set of two or more binding sites. These binding sites are the tile’s *inputs* (Figure 1a): the identities of the binding sites together determine which tile can attach at a given site. In theory, this simple mechanism is sufficient to produce arbitrarily complex shapes and patterns.^{9–12} In practice, however, self-assembly is stochastic, and unfavorable attachments of tiles with one or more incorrect or absent inputs also occur.^{9,19} While tiles that attach unfavorably usually fall off quickly, occasionally such a tile is locked in by the subsequent favorable attachment of an adjacent tile, an event we will call an *insufficient attachment*. Because a tile that attaches unfavorably may not match some of its input binding sites, it may not be the correct tile in the desired pattern: an error has occurred. Subsequent algorithmic pattern formation can be severely disrupted, resulting in a grossly malformed product.

Figure 1a illustrates correct and erroneous assembly steps. Insufficient attachments at sites where a correct tile could have attached are called *growth errors*; they involve both a correctly matching binding site and a mismatch. Insufficient attachments on facets involve no mismatches; nonetheless,

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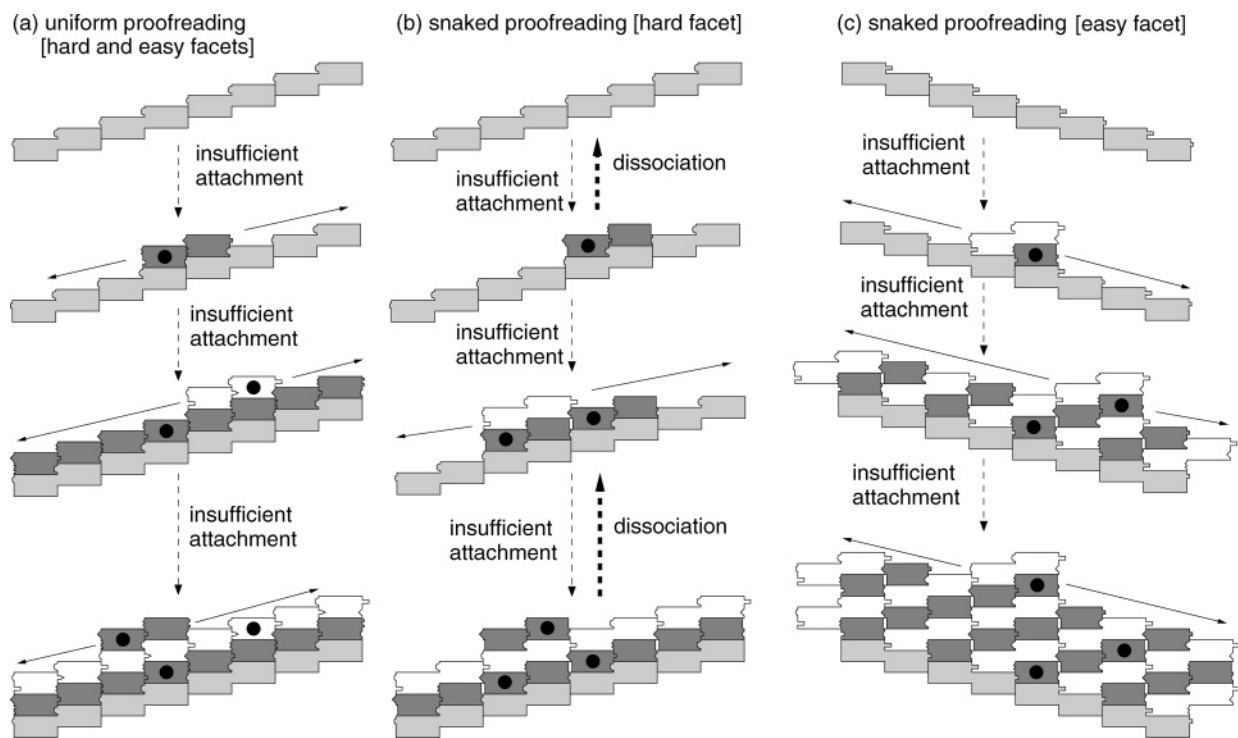


Figure 2. Facet nucleation and growth. (a) Facet nucleation of uniform proofreading blocks. Following a single insufficient attachment (tiles with dots indicate the unfavorable attachments that were locked in), subsequent growth by favorable attachments can grow an entire layer of tiles. Subsequent rows are each nucleated by a single insufficient attachment event. (b) Facet nucleation for snaked proofreading blocks along the hard facet. Here, a single insufficient attachment results in a pair of tiles on the facet, but further favorable growth steps are impossible because of the inert bonds interior to the snaked blocks. Two adjacent insufficient attachments are necessary to nucleate two layers of facet growth. Each additional two layers of growth requires another two adjacent insufficient attachments. (c) Facet nucleation for snaked proofreading blocks along the easy facet. Here, an insufficient attachment consists of a single tile and an adjacent double tile. Thus, two layers of snaked proofreading tiles can be nucleated by just one insufficient attachment.

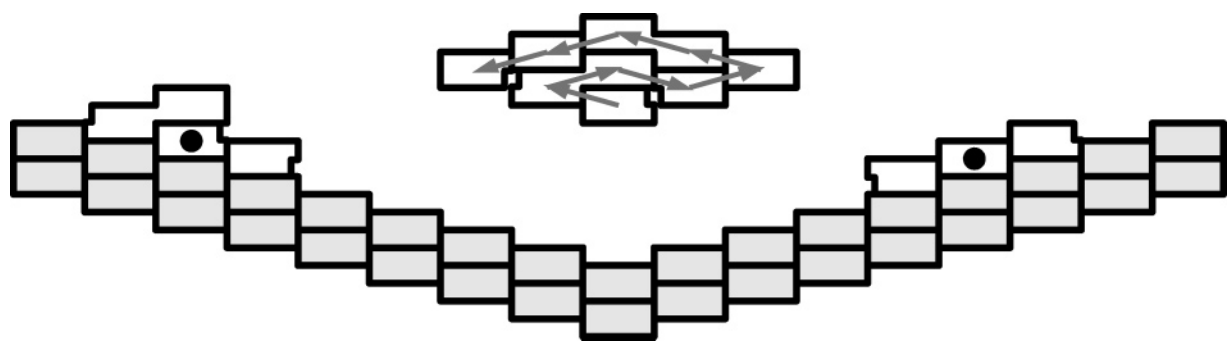


Figure 3. 3×3 snaked blocks reduce facet growth on both facet types. On either facet, an isolated insufficient attachment (initiated by the tiles marked with dots) can grow by favorable attachment to a maximal size of three tiles, at which point it is no longer possible to attach a tile by two binding sites. However, at a proper growth site, the series of exclusively favorable assembly steps following the snaked path shown can complete the block quickly.

the same rate as for the uniform proofreading tile set (Figure 2c).

The larger 3×3 snaked proofreading block (Figure 3) can protect against facet errors on both facet orientations, as is necessary for the full range of algorithmic growth. Simulation and theory predict that these and larger $k \times k$ blocks further reduce both growth errors and facet errors.²¹ The larger blocks use the basic mechanism of the 2×2 block multiple times to reduce the rate of facet errors; for example, the 3×3 block uses the 2×2 snaked motif once for each facet orientation. Thus, experimental investigation

of the 2×2 system assesses the essential principle used by the larger systems.

In this paper, we investigate experimentally whether 2×2 snaked proofreading tiles have a lower rate of facet nucleation than 2×2 uniform proofreading tiles. We use DNA tiles (Figure 4) to implement both uniform and snaked proofreading blocks and study their growth on long facets created using zigzag ribbons²⁴ (Figure 1d). See Supplementary Figures S1–S11 for sequences and diagrams of all molecules used in this work. We show that with snaked proofreading blocks, facet nucleation errors are reduced

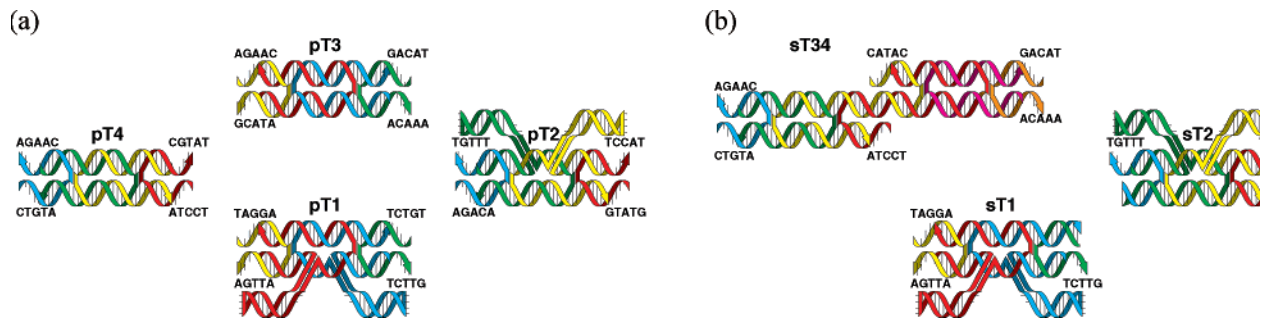


Figure 4. DNA implementation of uniform and snaked proofreading blocks. (a) DNA tiles for a uniform proofreading block. Each tile shown is a double-crossover molecule known as the DAO-E molecule.²⁵ A DAO-E molecule is composed of four strands of DNA. While the “core” of the molecule is double stranded, there are four five-nucleotide single-stranded regions (sticky ends) on each molecule that can bind to complementary sticky ends on other tiles. Two hairpins are present on each of the shaded tiles in Figure 1b to provide AFM contrast. (b) DNA tiles for a 2×2 snaked proofreading block, which in the rest of the paper will be referred to simply as a “snaked proofreading block”. To make an inert bond between sT1 and sT2, the sticky ends of the tiles are double stranded and truncated by two and three bases, respectively. The double tile sT34 is implemented by a larger molecule which has the structure of two DAO-E tiles fused together. Hairpins are used on tiles shaded in Figure 1c.

significantly on the expected facet. This demonstrates the effectiveness of the basic principle underlying snaked proofreading.

Results. (A) Lattice Structure. We first verified that the DNA structures we designed to implement uniform proofreading, snaked proofreading, and facets formed as expected. We used gel electrophoresis to verify that each of the four tiles in both proofreading tile sets assembled correctly. The component strands for each tile were annealed in a PCR machine (Eppendorf) from 90 to 20 °C at a rate of 0.1 °C every 6 s. Nondenaturing gel electrophoresis of the annealed tiles showed that the strands for each of the tiles in Figure 4 formed a single product with at least 80% yield. To make lattices, we mixed the annealed tiles for each lattice at room temperature to a final concentration of 100 nM per strand. We waited for half an hour and then deposited the samples onto mica and imaged by tapping mode atomic force microscopy (AFM). Both the uniform and proofreading tiles formed lattices (Figure 5a,c). (Compared to simply annealing all strands together in one step, this procedure is known to result in smaller and poorly formed crystals, but it is closer to the conditions used in experiments to measure facet nucleation.) The bright stripes in every other row of tiles confirmed that the tiles in the lattices were arranged correctly.

We determined the annealing and melting temperatures of each lattice using measurements of 260 nm absorbance²⁶ made by a spectrophotometer with a computer-controlled temperature bath (AVIV Biomedical) during an anneal and subsequent melt. The strands for each lattice (50 nM per strand) were annealed from 90 to 10 °C over 10 h, held at 10 °C for 2 h, and then melted back to 90 °C at the same rate. For both lattices, we observed (Figure 12 in Supporting Information) a reversible transition between 45 and 70 °C, where tile formation has been observed previously,^{3,17,24} and a hysteretic transition at lower temperatures, where lattices formed. Formation of both lattices occurred around 16 °C and melted between 25 and 37 °C, indicating a significant kinetic barrier to homogeneous nucleation. As ribbons would lower this kinetic barrier, we expected that in the presence

of ribbons at room temperature, lattices would grow on ribbons rather than nucleating spontaneously. At lower temperatures during melting, the uniform proofreading lattice absorbance signal was very noisy (presumably due to light scattering that occurs when the lattices grow larger than the 260 nm wavelength being measured), but it appears that the uniform proofreading lattices melted at a temperature ~ 3 °C higher than the snaked proofreading lattices.

Having demonstrated that the DNA implementation of both proofreading blocks can form crystalline lattices, and are thus structurally sound for investigating their relative effectiveness for reducing assembly errors, we proceeded to construct four types of zigzag ribbon, each of which presents a different combination of one of three facet types: “easy,” where snaked proofreading tiles can nucleate growth with just one insufficient attachment; “hard,” where two adjacent insufficient attachments are required for snaked proofreading tiles to nucleate facet growth; or blunt, which contain no binding sites and therefore do not allow growth. The “ZZeasy” ribbon has sticky ends for the easy facet (E1* and E2*) on the Z78 tile and blunt sticky ends on the Z56 tile, while the “ZZhard” ribbon has sticky ends for the hard facet (H1* and H2*) on the Z78 tile and blunt sticky ends on the Z56 tile. To directly compare the rates of facet nucleation on the two facet types within the same experiment, we used ribbons with a distinct facet type on each side of the ribbon so that we could compare the number of layers that grow on each side. The “ZZeasyhard” ribbon presents the easy facet on the Z78 tile and the hard facet on the Z56 tile, while the “ZZhardeasy” ribbon presents the hard facet on the Z78 tile and the easy facet on the Z56 tile.

To assemble each of the four ribbon types used to create the desired facets, we annealed their component tiles as described above and then mixed them together at room temperature to a final concentration of 100 nM of each tile. We annealed this mixture from 60 to 20 °C with the temperature decreasing 0.1 °C per minute. AFM imaging after dilution to 10 nM showed that each ribbon type formed four-tile-wide zigzag ribbons with a typical length of several micrometers (Figure 5b).

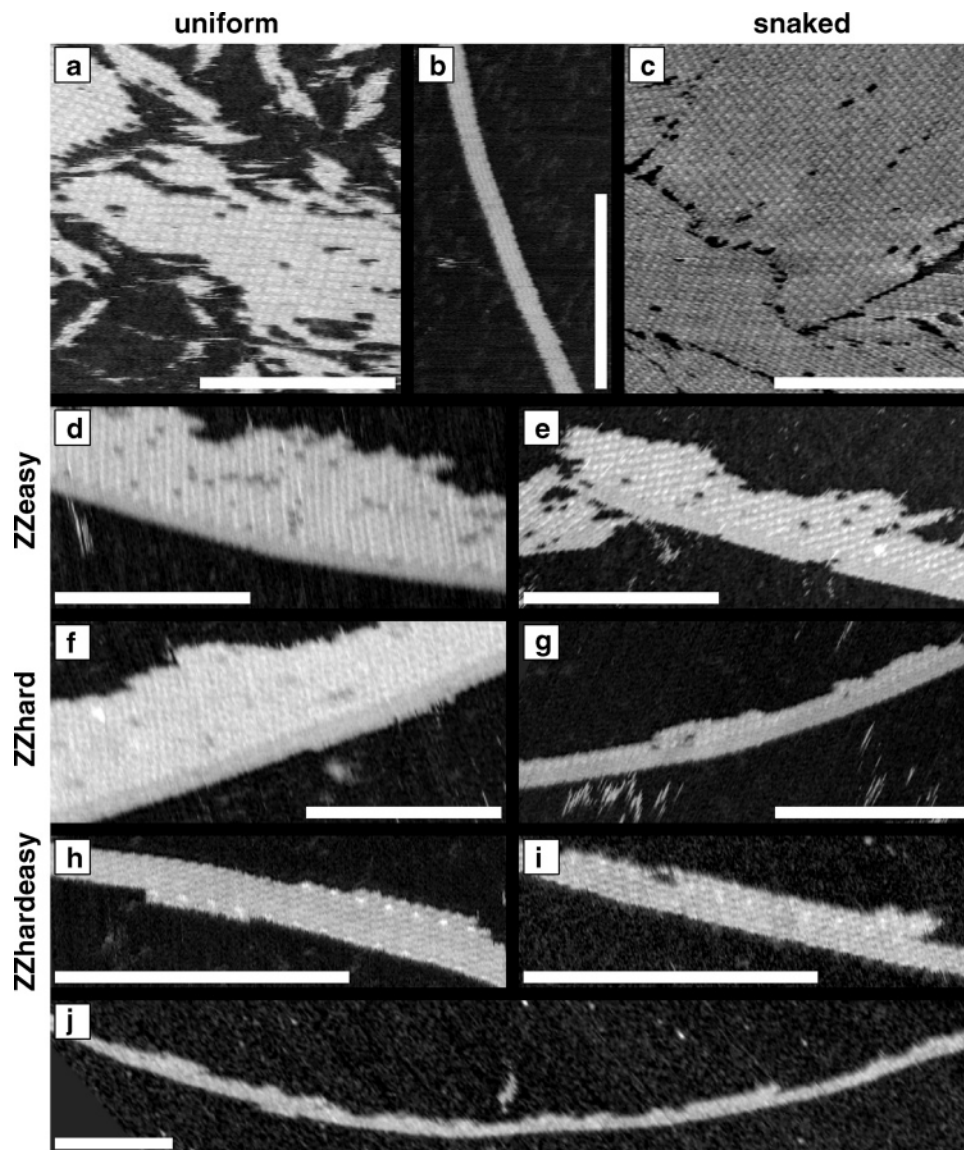


Figure 5. AFM images. Missing tiles are due to damage during AFM scanning. Scale bars are 300 nm. (a) Uniform proofreading lattices (100 nM). (b) Zigzag ribbons, ZZhardeasy (10 nM). (c) Snaked proofreading lattices (100 nM). (d) ZZeasy (10 nM) with uniform proofreading (50 nM). (e) ZZeasy (10 nM) with snaked proofreading (50 nM). (f) ZZhard (10 nM) with uniform proofreading (50 nM). (g) ZZhard (10 nM) with snaked proofreading (50 nM). (h) ZZhardeasy (10 nM) with uniform proofreading (10 nM). (i) ZZhardeasy (10 nM) with snaked proofreading (10 nM). (j) A long ZZhardeasy ribbon with snaked proofreading.

(B) Facet Growth on Single-Sided Ribbons. We used single-sided ribbons to measure the rate at which lattices grow on a facet. Since each row of growth must be nucleated by one or more insufficient attachments (Figure 2), the number of rows that grow on a facet in a fixed period of time increases with increasing nucleation rate. For each kind of proofreading lattice, we assembled ribbons that presented either an easy or hard facet as described above and then diluted them to 10 nM. We immediately added 50 nM of each preformed proofreading tile, waited for 10 min, then deposited the samples onto mica and imaged them using AFM.

Both kinds of proofreading lattices grew on both facets (parts d–g of Figure 5). The uniform proofreading lattices grew more than 10 layers on both facets, as did the snaked proofreading lattice on the easy facet. In contrast, the snaked

proofreading lattice grew only two to six rows on the hard facet during the experiment, suggesting that the nucleation rate of the snaked proofreading lattice on the facet is less than that of the other cases. However, because most of the tiles were used up during the experiment (50 nM is enough to grow an average of 10 rows on each ribbon), it was not possible to quantify how much smaller the nucleation rate of snaked proofreading lattices on the hard facet is than the other rates. Reliability was also limited by trial-to-trial variations in experiment timing and in tile stoichiometry for both the lattices and the ribbons and by concerns about the difference in the melting temperatures of the two lattices.

(C) Facet Growth on Double-Sided Ribbons. We therefore devised a second set of experiments to quantitatively compare the relative growth rates of the two kinds of proofreading lattices on both easy and hard facets. In these

Table 1. Ratio of the Number of Tiles (or Number of Groups of Tiles) Attached on the Easy Facet to the Number of Tiles Attached on the Hard Facet, As Measured from the AFM Data^a

easy/hard ratio	ZZhardeasy	ZZeasyhard
uniform proofreading	tiles: 1.0 ± 0.2	tiles: 1.2 ± 0.2
	groups: 1.1 ± 0.2	groups: 1.5 ± 0.3
snaked proofreading	tiles: 4.8 ± 1.0	tiles: 3.9 ± 0.9
	groups: 4.8 ± 1.0	groups: 4.2 ± 0.9

^aError bars indicate 1 standard deviation and are calculated using bootstrapping.

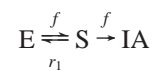
experiments, we measured the growth rate of the lattices on zigzag ribbons that present an easy facet on one side and a hard facet on the other. Because growth on both kinds of facets occurs under exactly the same physical conditions with identical thermodynamics on both sides, the ratio of growth on the two facets accurately measures the preference for growth on one facet over another. Growth on the easy and hard facet can be distinguished under the AFM because the orientation of stripes in the lattice with respect to the ribbon is different on the two sides (parts d–g of Figure 5).

For each lattice type, we performed experiments with double-sided ribbons using the same procedure as for the single-sided zigzag ribbons above. Experiments where we added 50 nM of proofreading tiles to the ZZeasyhard ribbons produced aggregations of ribbons in which lattices growing off each side merged with those of other ribbons, and it was not possible to measure the relative growth rate of tiles on the two facets. Aggregation was not a problem when we used a lower concentration, 10 nM, of proofreading tiles. To ensure that unknown structural differences between the Z78 and Z56 tiles used on either side of the ribbons did not affect the attachment rate, we repeated the competition experiment with both kinds of proofreading tiles at 10 nM with double-sided ribbons where the facet types were placed on opposite sides of the ribbon (ZZhardeasy). For each of the four experiments, we counted the number of tiles and groups that grew on each side of approximately 25 μm of ribbon (parts h–j of Figure 5). (A group is a connected assembly of two rows of tiles that could have grown as a result of a single nucleation event.) The results (Table 1) quantitatively confirm the conclusions of the earlier experiment. Only snaked proofreading tiles prefer to attach to the easy facet. In both experiments with snaked proofreading tiles, there were at least four times more snaked proofreading tiles attached to the easy facet.

(D) Analysis. These experimental results may be compared with a kinetic Monte Carlo model of DNA tile crystal growth, the kinetic Tile Assembly Model (kTAM).⁹ In this model, a tile bound by a single sticky end will dissociate with rate $r_1 = k_f e^{\Delta G_1/RT}$, a tile bound by two sticky ends will dissociate with rate $r_2 = k_f e^{\Delta G_2/RT}$, while tiles arrive at a given site at rate $f = k_f C$, where C is the concentrations of free monomer tiles in solution. For the tiles used here, k_f , ΔG_1° , and ΔG_2° are not known but can be estimated based on the sticky end sequences, a nearest neighbor model of DNA thermodynamics,²⁷ and the kinetics of oligonucleotide hybridization.²⁸ We ran kTAM simulations of snaked tile

growth on easy and hard facets during 10 min with $C = 10$ nM, $\Delta G_1^\circ = \Delta G_{se}^\circ + \Delta G_{init}^\circ$ and $\Delta G_2^\circ = 2\Delta G_{se}^\circ + \Delta G_{init}^\circ$ with $\Delta G_{se}^\circ = -10.3$ kcal/mol and an entropic $\Delta G_{init}^\circ = 1.8$ kcal/mol at 25 °C, and $k_f = 17 \times 10^6$ /M/s (chosen to obtain a comparable total extent of growth within 10 min). The simulated easy/hard ratio was 5.7 ± 0.6 . While this is remarkable agreement with our data, the simulation results are quite sensitive to ΔG_{se}° and ΔG_{init}° , which we do not know with confidence. (The nucleation rate ratio is relatively unaffected by k_f .)

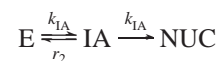
The simulations suggest that much better ratios could be achieved at higher temperatures, closer to T_m for the lattice, as can be understood using a simplified analytic model. The rate of an insufficient attachment, which is sufficient for nucleation with uniform proofreading and on the easy facet with snaked proofreading, is approximated as a two-step process²¹



where E is an empty site on a facet, S indicates attachment of a single isolated tile, and IA indicates an insufficient attachment has occurred. Thus,

$$k_{IA} \approx \frac{f(f+r_1)}{(f+(f+r_1))(f+r_1)}$$

where the first term is the overall rate for the two steps $E \rightarrow S$ followed by $S \rightarrow IA$ or $S \rightarrow E$, and the second term is the probability that IA was reached in that cycle. Similarly, on the hard facet with snaked proofreading, nucleation takes place when two insufficient attachments occur next to each other:



yielding

$$k_{NUC} \approx \frac{k_{IA}(k_{IA}+r_2)}{k_{IA}+(k_{IA}+r_2)} \frac{k_{IA}}{(k_{IA}+r_2)}$$

Near T_m , where $f \approx r_2$, we have $r_1 \gg f$ and $k_{IA} \approx f(f/r_1) \approx f e^{(\Delta G_2^\circ - \Delta G_1^\circ)/RT} = f e^{\Delta G_{se}^\circ/RT}$. Thus, $r_2 \gg k_{IA}$ so $k_{NUC} \approx k_{IA}(k_{IA}/r_2)$. Consequently, the ratio of nucleation rates on the easy and hard facets is approximately $k_{IA}/k_{NUC} \approx r_2/k_{IA} \approx e^{-\Delta G_{se}^\circ/RT}$. For a concentration $C = 10$ nM, at T_m we have $\Delta G_{se}^\circ \approx -6.4$ kcal/mol (as calculated based on $f \approx r_2$ and hence $\Delta G_2^\circ = RT \ln C$; the sticky end bonds will be weaker than those at 25 °C), but this still provides a remarkably favorable ratio of nucleation rates. By contrast, without the simplifications that apply near T_m , these formulas predict that the ratio decreases quickly as sticky end bonds become stronger, dropping to moderate values as the experimental conditions are approached. (This simplified theory predicts lower nucleation rates than those obtained in the simulations, however.)

For a tile set to be effective at reducing error rates in algorithmic self-assembly, not only must it reduce facet nucleation errors and growth errors, but proper growth must proceed unhindered. In kTAM simulations, both proofreading tile sets grow at roughly the same rate at growth sites, since there is always a location where a tile can attach by two binding sites. According to the simplified theory, growth at proper attachment sites, $k_{\text{GROW}} = f - r_2$, also decreases as T_m is approached, but k_{IA} and k_{NUC} decrease much more quickly, so that near T_m gives the most favorable ratio of growth to facet nucleation.

The use of these models for quantitative predictions is limited by our lack of accurate parameters for the full kTAM and by the nature of approximations used in the simplified analytic model. For example, ΔG_1° could be different for each sticky end sequence and ΔG_2° might not be a simple sum if steric effects come into play. Furthermore, it is possible that the favorable energy of blunt end stacking²⁹ between sT1 and sT2 needs to be taken into account; our model neglects it. There may also be small but nonzero energies between nonmatching sticky ends as well as stoichiometry imbalance among the tile types. Errors in the simplified analytic model, relative to the kTAM, arise due to the complexities of the crystal growth process, such as the combinatorial multiplicity of growth pathways, that are difficult to incorporate in a simple theory. Despite these caveats, we believe that the simulation and model provide valuable insights that can explain and guide experimental studies, for example, by suggesting how to change experimental conditions to obtain better performance.

Conclusions. We have shown here that snaked proofreading tiles reduce the rate of tile growth on long facets by at least a factor of 4 at 25 °C. Thus, we conclude that the snaked proofreading blocks reduce facet nucleation as designed, confirming the soundness of the basic principle of snaked proofreading. Presuming that growth at proper growth sites proceeds at comparable rates for both tile sets, which is not inconsistent with the extent of growth observed on ribbons, this suggests that in algorithmic growth the rate of errors per tile should be reduced. An important future experiment would be to verify this prediction. Another important direction is to determine how much error rates improve at optimal algorithmic assembly conditions, close to the melting temperature of the lattices. More generally, however, these results suggest that logical redesign of a set of tiles can reduce errors and that control over the growth path for self-assembly can be used to selectively inhibit undesired assembly processes. The general concept of robust proofreading tile sets is therefore worthy of further experimental and theoretical study.

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Supporting Information Available: Description of experimental methods, diagrams for the DX molecules used,

sequences for all DNA strands, and UV melting data. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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